

15 μm 128 x 128 GaAs/Al_xGa_{1-x}As Quantum Well Infrared Photodetector Focal Plane Array Camera

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ABSTRACT

In this paper, we discuss the development of very sensitive very long wavelength infrared GaAs/Al_xGa_{1-x}As quantum well infrared photodetectors (QWIPs) based on bound-to-quasibound intersubband transition, fabrication of random reflectors for efficient light coupling, and the demonstration of a 15 μm cutoff 128 x 128 focal plane array imaging camera. Excellent imagery, with a noise equivalent differential temperature (NE \cdot T) of 30 mK has been achieved.

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Introduction

There are several applications that require very long wavelength, large, uniform, reproducible, low cost, low 1/f noise, low power dissipation, and radiation hard infrared (IR) focal plane arrays (FPAs). For example, the absorption lines of many gas molecules, such as ozone, water, carbon monoxide, carbon dioxide, and nitrous oxide occur in the wavelength region from 3 to 18 μm . Thus, IR imaging systems that operate in the very long wavelength IR (VWIR) region (12 - 18 μm) are also required in many space applications such as monitoring the global atmospheric temperature profiles, relative humidity profiles, cloud characteristics, and the distribution of minor constituents in the atmosphere which are being planned for NASA's Earth Observing System [1]. In addition, 12-18 μm FPAs would be very useful in detecting cold objects such as ballistic missiles in midcourse (when hot rocket engine is not burning most of the emission peaks are in the 8-15 μm IR region) [2]. The GaAs based Quantum Well Infrared Photodetector (QWIP) [3,4,5] is a potential candidate for such space borne applications and it can meet all of the requirements mentioned above for this spectral region.

Fig. 1 shows the schematic conduction band diagram of a typical *bound-to-continuum* QWIP [6] which utilizes bound-to-continuum intersubband absorption. By carefully designing the quantum well structure, as well as the light coupling to the detector, it is possible to optimize the material to have an optical response in the desired spectral range and determine the spectral response shape [7]. In QWIPs, the dark current originates from three different mechanisms [8]. As shown in Fig. 1, the dark current arising from the first process is due to quantum mechanical tunneling from well to well through the $\text{Al}_x\text{Ga}_{1-x}\text{As}$ barriers (sequential tunneling). This process is independent of temperature. Sequential tunneling dominates the dark current at very low temperatures (<30 K for 15 μm QWIPs). The second mechanism is thermally assisted tunneling which involves a thermal excitation and tunneling through the tip of the barrier into the continuum energy levels. This process governs the dark current at medium temperatures.

The third mechanism is classical thermionic emission and it dominates the dark current at higher temperatures (>45 K for $15\ \mu\text{m}$ QWIPs). Consequently, for QWIPs operating at higher temperatures the last mechanism is the major source of dark current [8]. Therefore, the VWIR FPA we have discussed here in detail consisted of *bound-to-quasibound* QWIPs [9]. The advantage of the bound-to-quasibound QWIP over the bound-to-continuum QWIP [9] is that in the case of bound-to-quasibound QWIP the energy barrier for thermionic emission is the same as it is for photoionization as shown in Fig. 2. In the case of a bound-to-continuum QWIP the energy barrier for the thermionic emission is 10 - 15 meV less than the photoionization energy. Thus, the dark current of bound-to-quasibound QWIPs is reduced by an order of magnitude (i.e., $I_d \propto e^{-\frac{\Delta E}{kT}} \approx e^{-2}$ for $T = 55$ K).

Test Structure Results

The device structure consists of 50 periods containing $65\ \text{\AA}$ wells of GaAs (doped $n = 2 \times 10^{17}\ \text{cm}^{-3}$) and $600\ \text{\AA}$ barriers of $\text{Al}_{0.15}\text{Ga}_{0.85}\text{As}$ (sandwiched between $0.5\ \mu\text{m}$ GaAs top and bottom contact layers doped $n = 2 \times 10^{17}\ \text{cm}^{-3}$) grown on a semi-insulating GaAs substrate by molecular beam epitaxy (MBE). Then a $1.1\ \mu\text{m}$ thick GaAs cap layer on top of $300\ \text{\AA}$ $\text{Al}_{0.15}\text{Ga}_{0.85}\text{As}$ stop-etch layer was grown *in situ* on top of the device structure to fabricate the light coupling optical cavity. The MBE grown QWIP structure was processed into $200\ \mu\text{m}$ diameter mesa test structures (area = $3.14 \times 10^{-4}\ \text{cm}^2$) using wet chemical etching, and Au/Ge ohmic contacts were evaporated onto the top and bottom contact layers. The dark current-voltage curves of the QWIP was measured as a function of temperature from $T = 30$ - 60 K and the $T = 55$ K curve is shown in Fig. 2. The asymmetry in the dark current is attributed to the Si dopant migration into the growth direction [10] and hence higher asymmetry in the band structure.

The responsivity spectra of these detectors were measured using a $1000\ \text{K}$ blackbody source and a grating monochromator. The absolute peak responsivities (R_p) of the detectors were measured using a calibrated blackbody source. The detector were back

illuminated through a 45° polished facet [7] and its responsivity spectrum is shown in Fig. 3. The responsivity of the detector peak at 14.2 μm and the peak responsivity (R_p) of the detector is 420 mA/W. The spectral width and the cutoff wavelength are $\Delta\lambda/\lambda = 13\%$ and $\lambda_c = 14.9 \mu\text{m}$. The bias dependent peak responsivity of the detector is shown in Fig. 4. The measured absolute responsivity of the detector is small up to about $V_B = 1 \text{ V}$. Beyond that it increases nearly linearly with the bias reaching $R_p = 560 \text{ mA/W}$ at $V_B = 4 \text{ V}$. This type of behavior of responsivity versus bias is typical for a bound-to-quasibound QWIP. The peak quantum efficiency was 3% (lower quantum efficiency is due to the lower well doping density) for a 45° double pass.

The current noise i_n was measured using a spectrum analyzer and experimentally determined the photoconductive gain [11] g using $g = i_n^2 / 4eI_D B + 1/2N$, where B is the measurement band width and N is the number of quantum wells. As shown in Fig. 5, photoconductive gain of the detector reached 0.32 at $V_B = -6 \text{ V}$. Since the gain of QWIP is inversely proportional to the number of quantum wells N , the better comparison would be the well capture probability p_c , which is directly related to the gain [11] by $g = 1/Np_c$. The calculated well capture probabilities are 20% at low bias and 6% at high bias voltage which indicate the excellent hot-electron transport in this device structure. The peak detectivity is defined as $D_p^* = R_p \sqrt{AB} / i_n$, where R_p is the peak responsivity, A is the area of the detector and $A = 3.14 \times 10^{-4} \text{ cm}^2$. The measured peak detectivity at bias $V_B = -3 \text{ V}$ and temperature $T = 55 \text{ K}$ is $1.6 \times 10^{10} \text{ cm}^2/\text{Hz/W}$.

Light Coupling

QWIPs do not absorb radiation incident normal to the surface since the light polarization must have an electric field component normal to the superlattice (growth direction) to be absorbed by the confined carriers. As shown in Fig. 6(a) when the incoming light contains no polarization component along the growth direction, the matrix element of the interaction vanishes (i.e., $\vec{\epsilon} \cdot \vec{p}_z = 0$ where $\vec{\epsilon}$ is the polarization and \vec{p}_z is the

momentum along the z direction). As a consequence, these detectors have to be illuminated through a 45° polished facet [7] as shown in Fig. 6(b). Clearly, this illumination scheme limits the configuration of detectors to linear arrays and single elements. For imaging, it is necessary to be able to couple light uniformly to two dimensional arrays of these detectors.

Many more passes of IR light inside the detector structure can be obtained by incorporating a randomly roughened reflecting surface on top of the detectors which also removes the light coupling limitations and makes two dimensional QWIP imaging arrays feasible. A factor of eight enhancement in QWIP responsivity compared to 45° illumination geometry has been achieved with a randomly roughened reflecting surface [12]. The random structure on top of the detector prevents the light from being diffracted normally backward after the second bounce as happens in the case of cross-grating. After each bounce, light is scattered at a different random angle and the only chance for light to escape out of the detector is when it is reflected towards the surface within the critical angle of the normal. For the GaAs/air interface this angle is about 17° , defining a very narrow escape cone for the trapped light. The reflector was designed with two levels of scattering surfaces located at quarter wavelength separations, as shown in Fig. 7. The area of the top unetched level is equal to the area of the etched level ($\lambda/4$ deep). Therefore, the normally reflected light intensities from the top and bottom surfaces of random reflector are equal and 180° out of phase, thus maximizing the destructive interference at normal reflection and hence lowering the light leakage through the escape cone. This random structure was fabricated on the detectors by using standard photolithography and CCl_2F_2 selective dry etching. The advantage of the photolithographic process over a completely random process is the ability to accurately control the feature size and preserve the pixel to pixel uniformity which is a prerequisite for high sensitivity imaging focal plane arrays. Unlike periodic gratings the light trapping efficiency of the random reflector is independent of the wavelength [12], as a consequence it preserves the spectral purity.

Imaging Arrays

The photoconductive QWIPs of the 128x128 FPAs were then fabricated by wet chemical etching through the photosensitive GaAs/Al_xGa_{1-x}As multi quantum well layers into the 0.5 μm thick doped GaAs contact layer. Figure 8 shows thirty five 128x128 QWIP focal plane arrays on a 3 in. GaAs wafer. The pitch of the FPA is 50 μm and the actual pixel size is 38x38 μm^2 . Then the random reflectors on the top of the detectors were covered with Au/Ge and Au for Ohmic contact and reflection. Then indium bumps were evaporated on top of the detectors for Si read out circuit (ROC) hybridization. A single QWIP FPA was chosen (cutoff wavelength of this sample is 14.9 μm) and bonded to a 128x128 Si multiplexer (Amber AE-159) and biased at $V_b = -2.7$ V. The FPA was back-illuminated through the flat thinned substrate (thickness ≈ 25 μm). Substrate thinning from 630 μm to 25 μm has tremendously reduced the optical cross-talk between neighboring pixels. However, thinning of the GaAs substrate to a membrane level will completely eliminate the optical cross-talk between pixels. Furthermore, it will enhance the light trapping efficiency of the random reflector and completely eliminate the thermal mismatch between the silicon readout multiplexer and the GaAs based QWIP focal plane array. This initial array gave excellent images with 99.9% of the pixels working, demonstrating the high yield of GaAs technology. Figure 9 clearly shows the excellent uncorrected photocurrent uniformity (pixel-to-pixel) of the 16384 pixels of the 128x128 FPA with a standard deviation of only $\sigma=2.4\%$. The residual non-uniformity after correction was 0.05% and it is comparable to other types of focal plane arrays. Measurements were made to evaluate the variation of residual non-uniformity after two point correction as a function of scene temperature. According to these experiments the residual non-uniformity is less than 0.1% from 10 to 40 Celsius. This is a further indication that correctability is independent of random reflector and pixel-to-pixel lithography variations. As mentioned earlier this high yield is due to the excellent GaAs growth uniformity and the mature GaAs processing technology.

Video images were taken at various frame rates varying from 50 to 200 Hz with f/2.3 KRS-5 optics at temperatures as high as $T = 45$ K, using a ROC capacitor having a charge capacity of 4×10^7 electrons. Figure 10 shows an image of a man's face taken with

a QWIP FPA camera. The diffraction limited spot size of this 15 μm imaging camera is about 84 μm (i.e., $2.44.\lambda.f\#$). Therefore, the minimum spatial resolution is four pixels and most of the fuzziness in the picture is attributed to this diffraction limitation. The measured noise equivalent temperature difference $\text{NE}\cdot\text{T}$ of the FPA was 30 mK at an operating temperature of $T = 45$ K for 300 K background. This reasonably agrees with our calculated value of 10 mK. We have used the following equation to calculate the $\text{NE}\cdot\text{T}$ of the FPA.

$$\text{NE}\Delta T = \frac{\sqrt{AB}}{D_B^* (dP_B / dT) \sin^2(\theta / 2)} \quad (1)$$

where D_B^* is the blackbody detectivity, dP_B / dT is the derivative of the integrated blackbody power with respect to temperature, and θ is the field of view angle [i.e., $\sin^2(\theta/2) = (4f^2+1)^{-1}$, where f is the f number of the optical system]. No band pass filters were used and are unnecessary in QWIP camera systems because of the narrow spectral response of QWIPs. It should be noted that these initial unoptimized FPA results are far from optimum. The QWIP device structures was not optimized; the gratings were also not optimized for the maximum light coupling efficiency; no microlenses were used; no antireflection coatings were used; the substrate was not thinned enough (the hybrid was thinned to 25 μm , however, it was not sufficient to improve the light coupling efficiency to small pixel and in fact this explains the slightly higher measured $\text{NE}\cdot\text{T}$); and finally the multiplexer used was a photovoltaic InSb multiplexer which is certainly not optimized to supply the proper bias and impedance levels required by photoconductive QWIPs. Implementation of these improvements should significantly enhance the QWIP focal plane array operating temperatures (i.e., 77 K for 10 μm and 55 K for 15 μm).

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Figure Captions

Fig. 1 Schematic diagram of the conduction band in a bound-to-continuum QWIP with an electric field. Absorption of IR photons can photoexcite electrons from the ground state of the quantum well into the continuum, causing a photocurrent. Three dark current mechanisms are also shown: ground state tunneling (1); thermally assisted tunneling (2); and thermionic emission (3).

Fig. 2 Comparison of dark currents of bound-to-continuum and bound-to-quasicontinuum VWIR QWIPs as a function of bias voltage at temperature $T = 55$ K.

Fig. 3 Responsivity spectra of a bound-to-quasicontinuum VWIR QWIP focal plane array sample at temperature $T = 55$ K. The spectral response peak at $14.2 \mu\text{m}$ and the long wavelength cutoff is at $14.9 \mu\text{m}$.

Fig. 4 Peak responsivity as a function of bias voltage at temperature $T = 55$ K.

Fig. 5 Photoconductive gain as a function of bias voltage at temperature $T = 55$ K.

Fig. 6 (a) Intersubband absorption process of QWIPs at normal incidence. (b) 45° polished facet light coupling.

Fig. 7 Expanded corner of 128×128 QWIP FPA shows two level random reflectors on individual QWIP pixels ($38 \times 38 \mu\text{m}^2$). The random reflectors increase the light coupling efficiency by a factor of eight when the substrate is thinned down to $\sim 1 \mu\text{m}$

Fig. 8 Thirty five 128×128 QWIP focal plane arrays on a 3 in. GaAs wafer.

Fig. 9 Photosignal histogram of the 16384 pixel of the 128x128 array showing a high uniform uncorrected standard deviation of only $\sigma = 2.4\%$.

Fig. 10 One frame from a 15 μm QWIP video image of a man's face with $\text{NE}\cdot\text{T} = 30$ mK.